

Lattice Peptide Simulations using the Wang-Landau Monte Carlo Method

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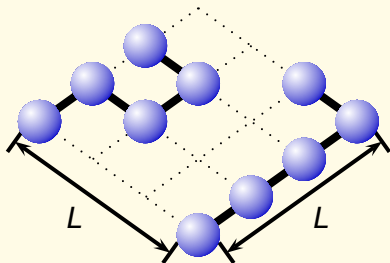


- 1 Lattice Polymers
- 2 Density of States Sampling
- 3 Results for Polymers and Peptides
- 4 Optimizing Wang-Landau
- 5 Knotted Ring Polymers

Simple Polymer Models



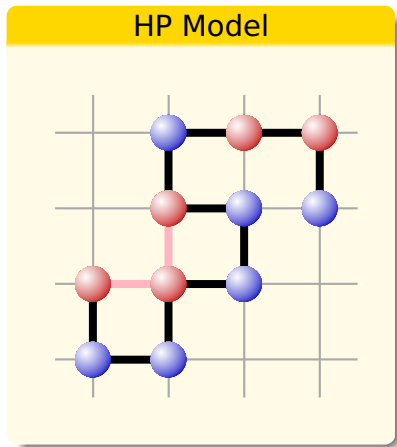
Lattice Polymer



- Defined on a lattice
- Sites may be occupied or unoccupied
- Nearest-neighbour interactions
- Connectivity defines polymer chains
- Chains may not cross

- Off-lattice versions are also commonly used.

The HP Model



- Self-avoiding chain of hydrophobic **H** and polar **P** residues,
- living on a 2D (square) or 3D (cubic) lattice.
- Each contact pair of non-bonded H residues contributes one unit $-\epsilon$ of favourable energy.
- Encapsulates the basic problems of folding.



K Lau, KA Dill, *Macromolecules*, **22**, 3986 (1989).

Folding Speed



- Proteins fold fast - sometimes in microseconds
 - Not by random searching of conformations!
 - How do they avoid the wrong routes?
 - “Levinthal paradox”
- Computer simulations need to be faster!

Dill has proposed a zipping-and-assembly model.

- local structures form in independent fragments
- then grow (zip) or coalesce with other structures (assemble) so as to optimise local contacts.

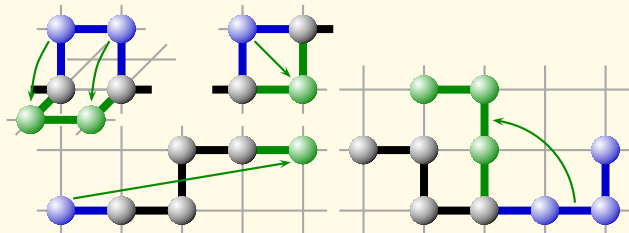


VA Voelz, KA Dill, *Proteins: Structure, Function, and Bioinformatics*, **66**, 877 (2007).

MC Moves for Lattice Polymers



Lattice Polymer Moves

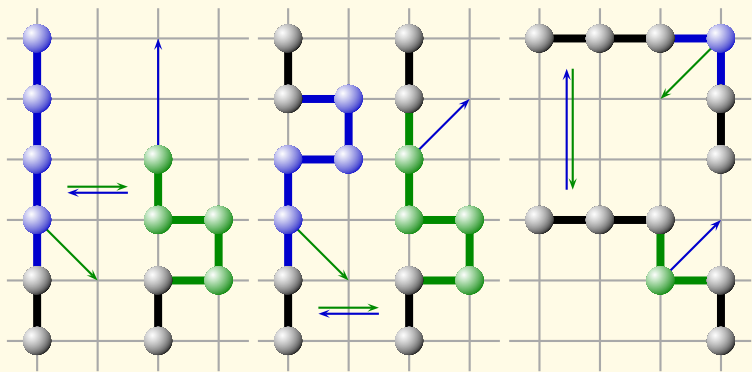


- Conventional lattice polymer Monte Carlo moves: crankshaft, corner flip, slithering snake, pivot.
- However, in dense phases, dramatic moves like pivot are likely to result in overlaps with surrounding atoms.

Novel Pull Moves



Defined by Single-Atom Move



Novel Pull Moves



- New pull moves allow local contacts to form.
- They are ergodic, and improve efficiency.
- Especially important when chain is closely packed.
- Include some conventional moves, e.g. corner-flip.
- This is a *biased* sampling method, and the way the moves are selected must be included in the acceptance/rejection criterion.
- Counting the available pull moves (forward and reverse) is a critical part of the method.



M Lesh, M Mitzenmacher, S Whitesides, *Proc. 7th Ann. Int. Conf. on Research in Computational Molecular Biology*, p188 (2003).



Novel Pull Moves

Pull Move Metropolis Equation

$$\mathcal{P}_{\text{acc}}(\mathbf{r}' \leftarrow \mathbf{r}) = \min \left(1, \frac{\alpha_{\text{pull}}(\mathbf{r} \leftarrow \mathbf{r}')}{\alpha_{\text{pull}}(\mathbf{r}' \leftarrow \mathbf{r})} e^{-\Delta E/k_B T} \right)$$

- ΔE is change in energy associated with $\mathbf{r}' \leftarrow \mathbf{r}$
- $\alpha_{\text{pull}}(\mathbf{r}' \leftarrow \mathbf{r}) = \mathcal{N}_{\text{pull}}(\mathbf{r}' \leftarrow \mathbf{r}) / \mathcal{N}_{\text{pull}}(\mathbf{r})$
 - $\mathcal{N}_{\text{pull}}(\mathbf{r}' \leftarrow \mathbf{r})$ = number of pull moves to \mathbf{r}' from \mathbf{r}
 - $\mathcal{N}_{\text{pull}}(\mathbf{r})$ = total number of pull moves from \mathbf{r}
- $\alpha_{\text{pull}}(\mathbf{r} \leftarrow \mathbf{r}') = \mathcal{N}_{\text{pull}}(\mathbf{r} \leftarrow \mathbf{r}') / \mathcal{N}_{\text{pull}}(\mathbf{r}')$ similarly.
- Need to consider both forward and reverse moves.
- Counting these available pull moves is quite time consuming.
- We improve the method by simplifying the counting (and rejecting some moves).



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Density of States Sampling



Efficient MC methods have been developed to map out behaviour of systems over a wide temperature range, including identification of low- E structures. With $\beta = 1/k_B T$, $\mathbb{W}(E)$ = density of states:

Thermodynamics from Density of States

partition function:
$$\mathbb{Q} = \sum_{\mathbf{r}} e^{-\beta E(\mathbf{r})} = \sum_E \mathbb{W}(E) e^{-\beta E}$$

average energy:
$$\langle E \rangle = \mathbb{Q}^{-1} \sum_E \mathbb{W}(E) E e^{-\beta E}$$

heat capacity:
$$C_V = (\langle E^2 \rangle - \langle E \rangle^2) / k_B T^2$$



Density of States Sampling

- $W(E)$ increases very rapidly with E .
- A **density-of-states simulation** samples states with a probability or weight $P(E) \propto 1/W(E)$.
- This covers the entire energy scale uniformly.
- Allows low-energy configurations to exchange with higher-energy ones and vice versa.

 FG Wang, DP Landau, *Phys. Rev. E*, **64**, 056101 (2001).

Acceptance-Rejection Criterion

$$P_{\text{acc}}(\mathbf{r}' \leftarrow \mathbf{r}) = \min \left(1, \frac{P(E(\mathbf{r}'))}{P(E(\mathbf{r}))} \right) = \min \left(1, \frac{W(E(\mathbf{r}))}{W(E(\mathbf{r}'))} \right)$$

Wang-Landau Sampling





$\mathbb{W}(E)$ is built up progressively during the simulation.

Wang-Landau Algorithm

- 1 Set all $\mathbb{W}(E) = 1$, set factor $f = e = 2.71828$.
- 2 Set all $h(E) = 0$.
- 3 Conduct MC moves using $\mathbb{W}(E)$, accepted states t
 - After each move $\mathbb{W}(E_t) \rightarrow f \times \mathbb{W}(E_t)$, $h(E_t) \rightarrow h(E_t) + 1$
- 4 Repeat from step 3 until $h(E)$ is “sufficiently flat”
- 5 $f \rightarrow \sqrt{f}$ (f decreases exponentially with time)
- 6 Repeat from step 2 until f is “sufficiently small”.

Hope $\mathbb{W}(E)$ converges to correct density of states.

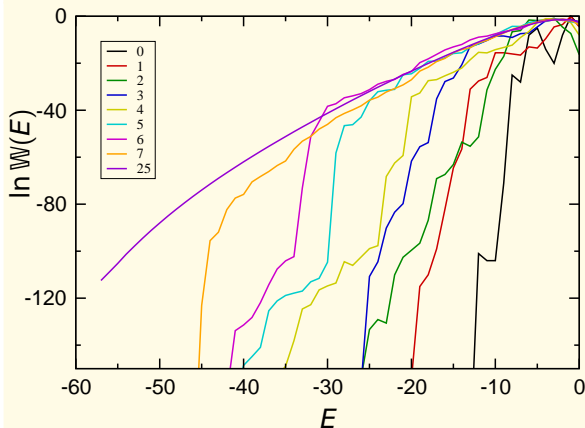


-  M Bachmann, W Janke, *J. Chem. Phys.*, **120**, 6779 (2004).
-  T Wüst, DP Landau, *Comp. Phys. Commun.*, **179**, 124 (2008).
- Wang-Landau covers the entire energy scale uniformly.
- Can map out behaviour of lattice polymers and peptides over a wide temperature range.
- Allows identification of low- E structures.

Pull Move Acceptance-Rejection Criterion

$$\mathcal{P}_{\text{acc}}(\mathbf{r}' \leftarrow \mathbf{r}) = \min \left(1, \frac{W(E(\mathbf{r}))}{W(E(\mathbf{r}'))} \frac{\alpha_{\text{pull}}(\mathbf{r} \leftarrow \mathbf{r}')}{\alpha_{\text{pull}}(\mathbf{r}' \leftarrow \mathbf{r})} \right)$$

Example: 103-bead Lattice Protein

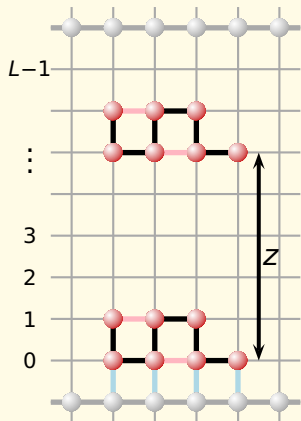
Successive f Iterations

“cytochrome C”
 PPHHPPPPPH...
 HPPHHPHPPH...
 PPPPPPHPP...
 PHHPHHPPPP...
 PPHPPHPHPP...
 HPPPPPHHHP...
 PPPHHPHHPP...
 PPPHHPPPPH...
 HHHPHPPPPP...
 PPPHHHHHPP...
 HPP

Polymers on Surfaces




Slit Geometry



- Confining wall at L
- Energy $E = -n\epsilon - s\sigma$
- $n = n(\mathbf{r}) =$ number of H-H contacts for internal configuration \mathbf{r} ;
- $-\epsilon$ = contact energy.
- $s = s(\mathbf{r}, z) =$ number of surface contacts for given \mathbf{r} , and position z ;
- $-\sigma$ = surface energy.
- Density of states $\mathbb{W}_L(n, s)$
- $\mathbb{Q} = \sum_n \sum_s \mathbb{W}_L(n, s) e^{+n\beta\epsilon + s\beta\sigma}$

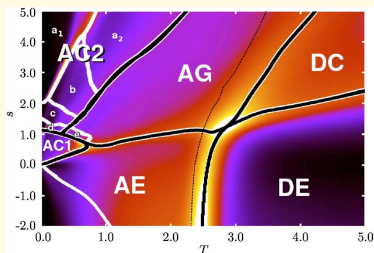
Polymers on Surfaces



 M Bachmann, W Janke, *Phys. Rev. E*, **73**, 020901 (2006); *ibid*, **73**, 041802 (2006).

- Studied H_{179} homopolymer in cavity with $L = 200$.
- Used chain-growth (PERM) rather than MC moves.
- Pseudo-phase diagram in T and $\chi = \epsilon/\sigma$.
- Transitions marked by peaks in “heat capacities”.

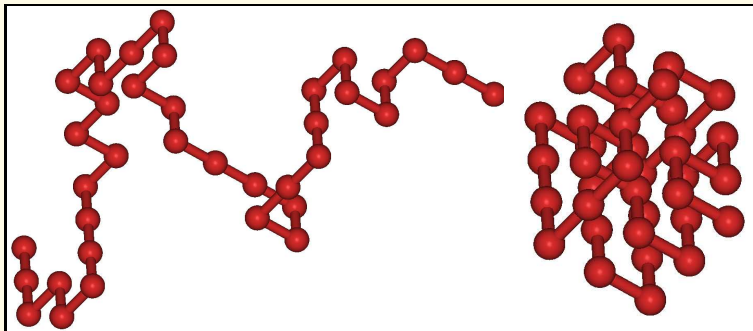
Phase Diagram



States Observed

- AE: Adsorbed-Expanded
- DE: Desorbed-Expanded
- AC: Adsorbed Compact
- DC: Desorbed Compact
- AG: Attached Globule

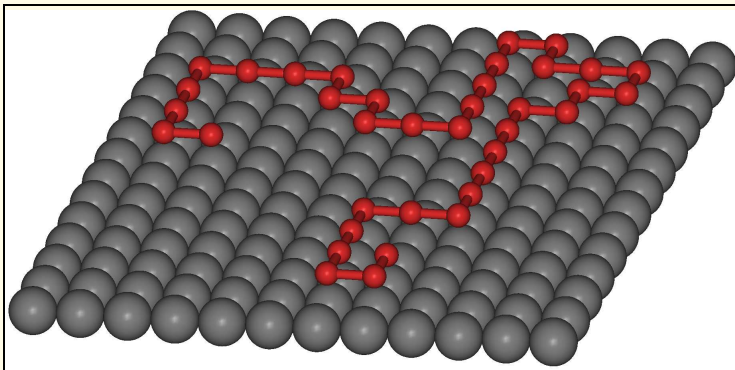
Desorbed Configurations



- Desorbed Expanded
- High T , low $\chi = \epsilon/\sigma$.

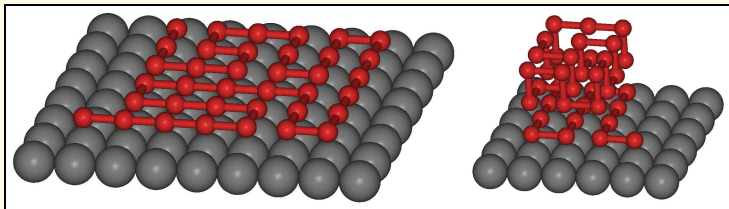
- Desorbed Compact
- High T , high $\chi = \epsilon/\sigma$.

Adsorbed Configurations



- Adsorbed Expanded
- Low T , low $\chi = \epsilon/\sigma$.

Adsorbed Configurations



- Adsorbed Compact
- Low T , high $\chi = \epsilon/\sigma$.
- Adsorbed Globule
- Medium T , high $\chi = \epsilon/\sigma$.



Polymers on Surfaces

Really interested in polymer adsorption on a **single** wall.
But cannot allow $L \rightarrow \infty$.

- Significant improvement of the method.
- There is no need for the second confining wall.

Redefine $s = s(\mathbf{r})$, as the number of beads on the lower bounding surface z of a polymer configuration \mathbf{r} .

New Surface Monte Carlo Algorithm

- 1 Standard pull move on isolated polymer
 - Count the monomer-monomer interactions n
- 2 Translate the surface to the plane of contact
 - Count the monomer-surface interactions s
- 3 Look up density of states $\mathbb{W}(n, s)$, accept or reject
- 4 NB: $\sum_s \mathbb{W}(n, s) = \mathbb{W}(n)$ for free polymer, as before.



Relate To Slit Geometry

$$W_L(n, s) = \begin{cases} AW(n, s) & s > 0 \\ A(L - D(n))W(n) & s = 0 \end{cases}$$

$$Q_L = A \sum_n \sum_{s>0} W(n, s) e^{+n\beta\epsilon} e^{+s\beta\sigma} + ALQ_{\text{int}} - A \sum_n W(n)D(n)e^{+n\beta\epsilon}$$

- A = cross-sectional area, L = wall separation.
- Q_{int} = internal partition function of free polymer.
- $D(n)$ is the average z -extent of polymers with n internal contacts

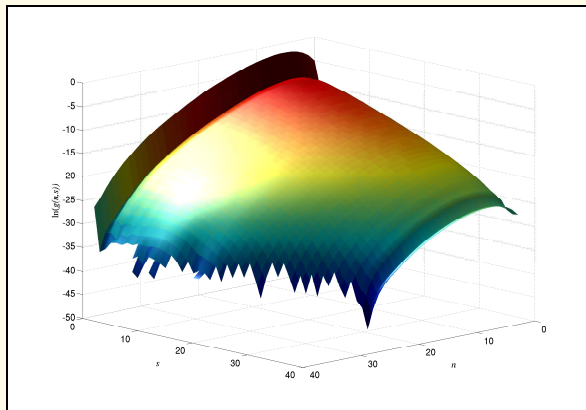


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H₃₇ Homopolymer



Density of States

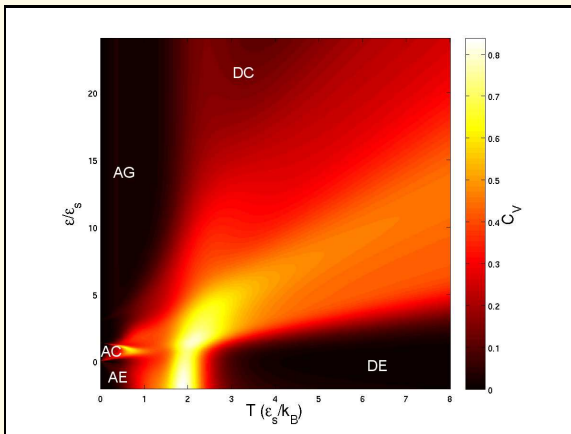


AD Swetnam,
MP Allen,
*Phys Chem
Chem Phys*,
11, 2046
(2009).

H₃₇ Homopolymer



Heat Capacities

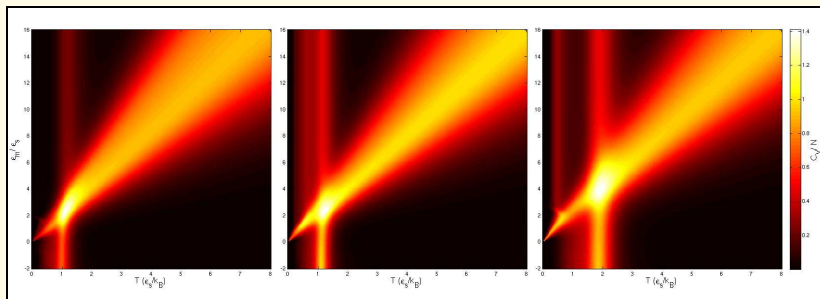


36-bead HP Peptide



PPPHHPPHHPPPPHHHHHHHPPHHPPPPHHPPHPP

Heat Capacities



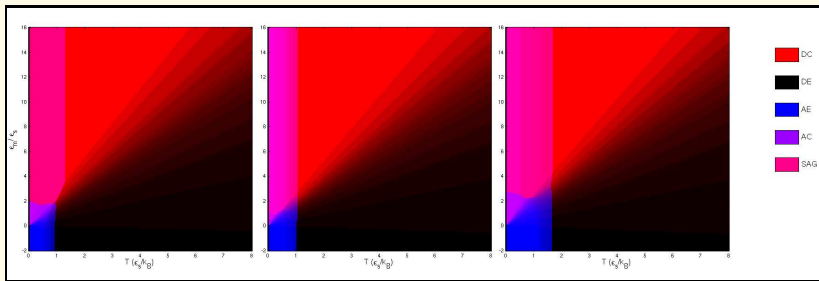
Left to right: surface attracts H only, P only, both H&P.
Pseudo-transitions indicated by regions of high C_V .

36-bead HP Peptide



PPPHPPHHPPPPPHHHHHHHHPPHHPPPPHHPPHPP

Phase Diagrams



Left to right: surface attracts H only, P only, both H&P.
 Phases are identified by monomer and surface contacts of free-energy-minimizing conformations.



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Optimizing Wang-Landau



Is it best to make f change exponentially?

- At each stage of Wang-Landau $f \rightarrow \sqrt{f}$.
- It may be better to make $\ln f \propto 1/t$

 RE Berardinelli, VD Pereyra, *J. Chem. Phys.*, **127**, 184105 (2007).

 C Zhou, J Su, *Phys. Rev. E*, **78**, 046705 (2008).

- To test properly requires exactly soluble model.
- Exact enumeration, isolated 18 bead homopolymer.
- Fractional error = $\epsilon_{\text{rms}} = \sqrt{\overline{\epsilon(E)^2}}$ where

$$\epsilon(E) = \frac{\ln W(E) - \ln W_{\text{exact}}(E)}{\ln W_{\text{exact}}(E)}$$



Conventional Wang-Landau

- Specify a flatness criterion: 80%... 99%
- When $\min(H) > \text{flatness} \times \bar{H}$, $f \rightarrow \sqrt{f}$.

Modified Wang-Landau

- Initially change modification factor in the normal way (in this case with flatness=80%)
- After the point where $\ln f < 1/t$, set $\ln f = 1/t$.

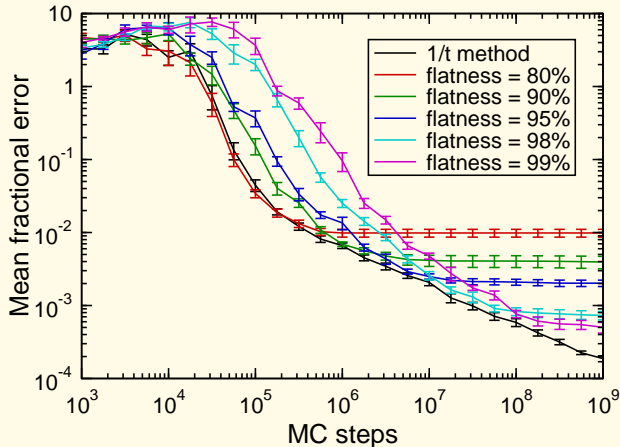
Here t is defined as the number of MC steps M divided by the number of energy levels N_E .

Optimizing Wang-Landau



10 runs were performed to get each set of data.

Comparison of Wang-Landau Methods



HP peptides are challenging!



- For HP sequences, some runs fail to converge.
- Adjust modification factor according to progress.
- Hard-to-access low-lying states
- Even the $1/t$ method has trouble

Proposed Improvement

$$\ln f = p \sum_E \left(\frac{H(E)}{\sum_E H(E)} - \frac{1}{N_E} \right)^2 = p \left(\frac{\sum_E H(E)^2}{M^2} - \frac{1}{N_E} \right)$$

p = adjustable prefactor.

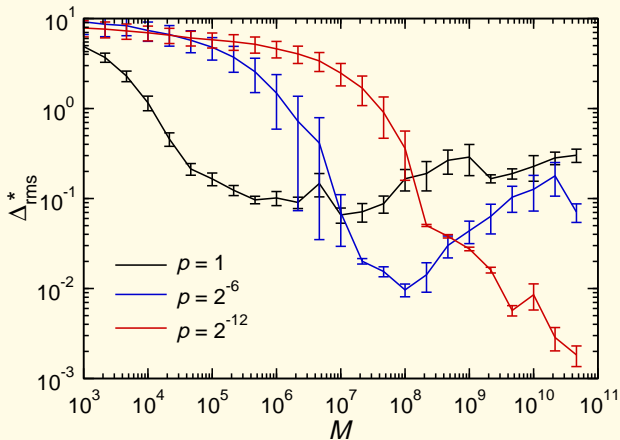


C Zhou, J Su, *Phys. Rev. E*, **78**, 046705 (2008).



AD Swetnam, MP Allen, *J. Comput. Chem.*, **32**, 816 (2011).

Example: 103-bead Lattice Protein

Various prefactors p 



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Ring Polymers

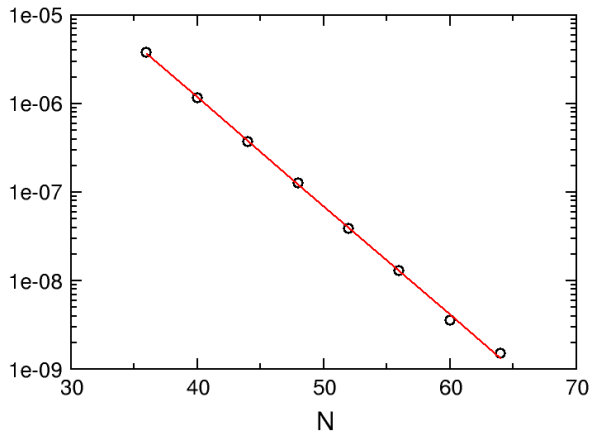


- Pull moves preserve knot topology.
- Wang-Landau sampling of $\mathbb{W}(E, L)$.
- $L = z$ -height (or wall separation).
- Plot C_V vs (T, F) , F is tensile/compressive force.
- Combinatorial effects diminish number of attempts
 $L = 2 \rightarrow L = 1$ for unknotted polymer and large N .
- Effectively a physical bottleneck in E, L -space.
- Sample the number of beads in each layer.

Ring Polymers



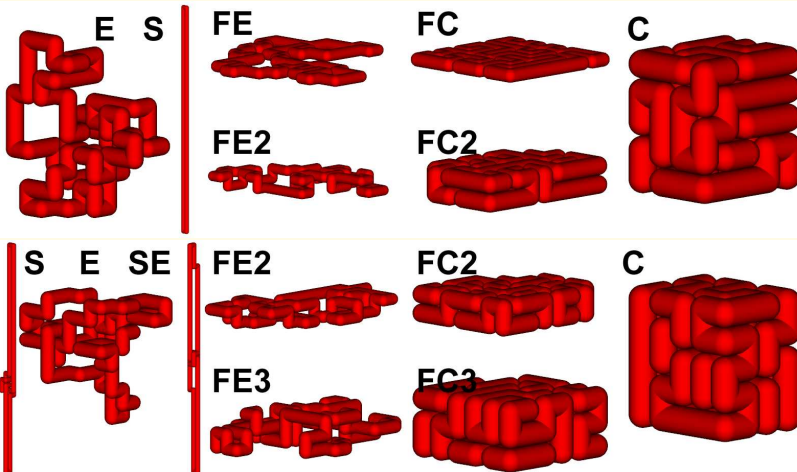
Attempt fraction for $L = 2 \rightarrow L = 1$ vs N



100-bead Ring Polymer



Un-knotted (above) and Trefoil-knotted (below)

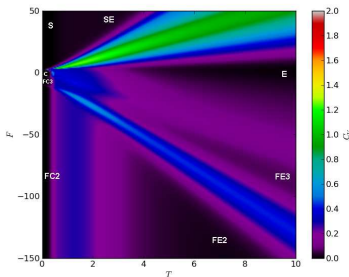
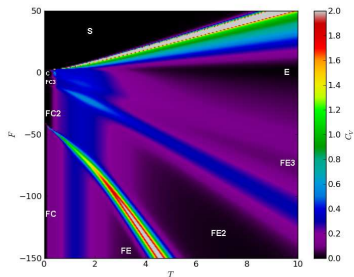


100-bead Ring Polymer



- Heat capacity maps (pseudophase diagrams).
- Strong transition to flat state in unknotted case.

Unknotted (left) and Trefoil-knotted (right)





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- Centre for Scientific Computing